

Article



A Comparison Study for Chloride-Binding Capacity between Alkali-Activated Fly Ash and Slag in the Use of Seawater

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Featured Application: The results of this study can be applied to the production of cementless concrete using seawater.

Abstract: This study aimed to investigate the chloride-binding capacity of alkali-activated fly ash (denoted as FSW) and slag (denoted as SSW) samples and their synthesized Cl-bearing phases, which are capable of binding and immobilizing chloride when seawater is used as the mixing water. This study also examined the progressive changes in the pore structures of the FSW and SSW samples over time. The results show that the SSW sample is significantly more effective in the uptake of chloride ions compared to the FSW sample at 28 days of curing. While the FSW sample forms Cl-bearing zeolites (Cl-chabazite and Cl-sodalite) (possibly with similar types of geopolymeric gels), the SSW sample synthesizes Cl-bearing, layered double hydroxides (LDH) (Cl-hydrocalumite and Cl-hydrotalcite). Although both samples involve Cl-binding phases, the FSW sample is likely to be less efficient because it largely produces zeolites (or similar geopolymeric gels) with no Cl-binding capability (i.e., zeolites X and Y). Meanwhile, the SSW sample produces Cl-bearing LDH phases as well as C-S-H(I), which can physically adsorb chloride. The SSW sample exhibits both pore-size refinement and porosity reduction over time, while the FSW sample only exhibits pore-size refinement. Therefore, the SSW system is more advantageous in the use of seawater because it more effectively prevents Cl ingression due to greater impermeability.

Keywords: alkali-activated binders; fly ash; ground-granulated blast furnace slag (GGBFS); seawater; chloride; zeolite; layered double hydroxide

1. Introduction

The global cement and concrete industry is facing environmental challenges. First, the calcination process for cement clinkers releases 0.8–1.0 tons of carbon dioxide to produce one ton of ordinary portland cement (OPC) [1,2]. As a result, the total CO₂ emission from worldwide OPC production is equivalent to 6–8% of global CO₂ emissions [3]. Furthermore, the production of OPC concrete consumes a significant quantity of fresh water, more specifically one billion tons of fresh water per year [4]. Currently, water shortages are already serious problems in the Middle Eastern and North African nations, which have already imported a huge volume of fresh water, such as Saudi Arabia, United Arab Emirates, Bahrain, Qatar, Kuwait, Yemen, Libya, Jordan, and Israel [5]. This water shortage may become a vital challenge to other nations in the near future due to climate change [6]. Finally, existing concrete structures suffer durability issues due to chemical deteriorations, such as chloride attacks, corrosion, and

sulfate attacks [2,7]. Therefore, the enhancement of concrete durability will help to improve environmental sustainability by reducing unnecessary resource consumption for maintenance.

New eco-efficient binders, such as calcium sulfoaluminate cements [8], super-sulfate cements [9], calcium aluminate cements [10], and alkali-activated binders [11,12], have been introduced and developed to address the significant CO_2 emissions from OPC production. The alkali-activated binder is likely to be a high-profile candidate to replace OPC due to its significantly low production of CO_2 and comparable performances of strength and durability to OPC [13].

To reduce the current level of fresh water consumption for OPC concrete production, many earlier studies have explored the possible use of seawater in the concrete's production [14,15]. However, the presence of chloride from seawater in concrete likely induces serious corrosion of embedded steel bars [2]. In the previous literature, alkali-activated binders showed better durability against a saline environment compared to OPC pastes [16]. Thus, in unavoidable situations leading to the use of seawater, the application of alkali-activated binders would be a better choice than OPC in terms of chloride-induced durability problems. In addition to using seawater, for concrete structures exposed to marine environments, alkali-activated binders could also provide a longer service life of concrete structures with better durability against seawater [17]. Consequently, alkali-activated binders may play important roles in addressing the aforementioned environmental concerns (i.e., CO_2 emissions, water shortages, and durability). Therefore, it is important to understand the interactions between seawater and alkali-activated binders.

Among the alkali-activated binders, two representatives are alkali-activated fly ash (also called fly ash-based geopolymer) and alkali-activated slag. In the previous literature, Arbi et al. well-reviewed the recent research on the chloride resistance of alkali-activated fly ash/slag systems [18], and showed that most studies on the chloride resistance of alkali-activated binders were based on measurements of rapid chloride permeability tests (RCPT) and non-steady-state migration (NSSM). Few studies have been conducted on Cl-bearing phases, which are capable of binding and immobilizing chloride, and stable phases when seawater is used as the mixing water in alkali-activated binders, while Friedel's salt and Kuzel's salt in OPC pastes have been studied actively [19,20]. Regarding the stable phases in alkali-activated fly ash with seawater, Belviso et al. attempted to synthesize zeolite from alkali-activated fly ash using seawater at various temperatures, and they found zeolite X, zeolite ZK-5, and sodalite in the samples [21]. Regarding Cl-bearing phases in alkali-activated slag, Khan et al. recently reported a formation of Friedel's salt in 5 M NaOH-activated slag with 2.47% NaCl solution [22]. Aside from these two studies, no study has compared alkali-activated slag and fly ash in terms of their Cl-binding capacities and their Cl-bearing phases.

Therefore, the present study aimed to fill the research gap by comparing the Cl-binding capabilities and Cl-bearing phases between alkali-activated slag and fly ash systems when seawater was used as a mixing water. The progressive change in pore structure over time, which is related to impermeability to Cl ingression, was also tested. A mixed solution of NaOH, Na₂SiO₃, and seawater was used as the main activating solution in this study because NaOH and Na₂SiO₃ have been often used as alkaline activators in alkali-activated binders [23]. Free and bound chloride contents were obtained by measuring acid-soluble and water-soluble chloride contents. Cl-bearing phases were identified through X-ray diffraction (XRD). Finally, mercury intrusion porosimetry (MIP) was conducted to observe changes in the pore structures. The results provide a better understanding of the Cl-bearing phases and pore structures in alkali-activated binders.

2. Experimental

2.1. Materials

Fly ash (Class F according to ASTM C 618 [24]) and ground-granulated blast furnace slag (GGBFS) were obtained. Table 1 lists the chemical compositions of the raw materials measured through X-ray fluorescence (XRF), and Figure 1 presents the XRD patterns of fly ash and GGBFS with the reference

peaks of the identified phases. Table 1 and Figure 1 show that both raw materials are general fly ash and GGBFS [25–27].



Figure 1. XRD patterns of (a) fly ash (FA) and (b) ground-granulated blast furnace slag (GGBFS).

A mixture of NaOH and Na₂SiO₃ (reagent grade, 10.6% Na₂O, 26.5% SiO₂, 62.9% H₂O, Sigma-Aldrich, St. Louis, MO, USA) was used as the alkaline activator. Ten moles (10 M) NaOH solution was made from NaOH pallet (reagent grade, \geq 98% purity) and the seawater.

The seawater was collected from Songdo in South Korea $(35^{\circ}04'32'' \text{ north}, 129^{\circ}01'01'' \text{ east})$. The pH value of the seawater was 8.4, and it contained several types of cations and anions. The cations $(Ca^{2+}, K^+, Mg^{2+}, \text{ and Na}^+)$ were measured using inductively coupled plasma atomic emission spectroscopy (ICP-AES), and the anions $(Cl^- \text{ and } SO_4^{2-})$ were quantified using ion chromatography (IC). Table 2 lists the chemical composition of the collected seawater and indicates that the used seawater had a typical chemical composition of seawater [28].

E	A	GG	BFS
Oxide	wt %	Oxide	wt %
SiO ₂	51.29	CaO	47.42
Al_2O_3	22.16	SiO ₂	31.97
Fe ₂ O ₃	9.71	Al_2O_3	12.51
CaO	6.44	MgO	3.21
K ₂ O	1.87	SO_3	2.51
Na ₂ O	1.73	TiO ₂	0.66
MgO	1.72	K ₂ O	0.53
TiO ₂	1.36	Fe ₂ O ₃	0.41
P_2O_5	1.29	MnO	0.34
SO_3	1.24	Na ₂ O	0.21
SrO	0.37	SrO	0.08
MoO ₃	0.34	BaO	0.07
BaO	0.23	ZrO_2	0.05
MnO	0.08	Cl	0.02
V_2O_5	0.05	-	-
ZnO	0.02	-	-
Cr_2O_3	0.02	-	-
CuO	0.02	-	-
NiO	0.02	-	-
Cl	0.01	-	-

Table 1. Chemical compositions of fly ash and ground-granulated blast furnace slag (GGBFS) used in this study.

Table 2. Ionic concentrations of seawater (ppm)
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Ca ²⁺	K+	Mg ²⁺	Na ⁺	Cl-	SO_4^{2-}
480	300	1200	10,000	15,000	2600

2.2. Sample Preparation

Fly ash and GGBFS were independently mixed with the alkaline activator. NaOH solution was prepared by dissolving NaOH pellets in seawater to 10 mol/L, and then cooling to room temperature. Na₂SiO₃ solution was mixed with NaOH solution with a weight ratio of 3:7 [29,30]. The weight ratio of the activator (10 M NaOH in seawater + Na₂SiO₃ solution) to the binder (fly ash or GGBFS) was fixed at 0.45. Table 3 summarizes the detailed mixture proportion of the samples. In the present study, the sample of alkali-activated fly ash with seawater (fly ash with seawater) will be denoted FSW and that of alkali-activated GGBFS (GGBFS with seawater) will be SSW in the following sections. After mixing, the pastes were cast into $50 \times 50 \times 50$ mm cubic molds for strength and XRD tests, and Φ 25.4 × 25.4 mm cylinder molds for MIP. The samples were cured at a temperature of 60 °C and a relative humidity of 99%. The compressive strengths of the hardened samples were measured at the age of 7 and 28 days of curing. The loading rate was 0.4 mm/min. Fractured specimens after a compressive strength test were finely powdered for the XRD and chloride tests. At the age of 7 and 28 days of curing, cross sections with a thickness of 2 mm along the length of the hardened cylindrical samples were prepared for MIP analysis using a precision saw. The specimens were examined by MIP.

Binder (g) (GGBFS or Fly Ash)	Activator	(g)	Activator/ Binder Ratio	Curing Temp (°C)
	10 M NaOH in Seawater	Na ₂ SiO ₃ Solution		
1600 (1240 or 1130 kg/m ³)	504	216	0.45	60

Table 3. Mix proportions of samples.

2.3. Test Methods

2.3.1. Compressive Strength Test

The compressive strengths of the hardened samples were measured at 7 and 28 days of curing. Figure 2 presents the test results as an average of the triplicate samples. Since the present study focused on the chloride-binding capabilities of the samples, the results of the strength tests are presented in this section. The strength development of the samples similarly followed the alkali-activated binders reported in previous studies [11–13,17,23,31]. The specimens that fractured after the compressive strength tests were finely ground for chloride content tests and XRD.



Figure 2. Compressive strengths of prepared samples of this study at 7 and 28 days. FSW: alkali-activated fly ash; SSW: alkali-activated GGBFS.

2.3.2. Measurement of Free and Bound Chloride Contents

Regardless of whether chloride ions originate from internal or external sources, chloride ions can be either free or bound chloride contents. Free chloride ions are presented in a free form in a pore solution, indicating that they can freely move through the pore solution in concrete, while bound chloride ions are immobilized chloride ions. The bound chloride contents can be calculated by subtracting a free chloride content from a total chloride content [32–34]. The free chloride and total chloride contents can be respectively measured from water-soluble (see ASTM C 1218 [35]) and acid-soluble (see ASTM C1152 [36]) chloride contents [35,36]. Even though the acid-soluble chloride contents depend on the solubility of Cl-bearing phases, ASTM C 1152 is the only well-established test method to measure the total chloride contents of cement-based samples.

2.3.3. XRD and MIP Tests

XRD was performed using a high-power X-ray diffractometer (D/Max2500V/PC, Rigaku, Tokyo, Japan) with a Cu K α radiation, at a scanning rate of 1°/min from 5 to 60° 2 θ . The XRD data was

analyzed using X'pert HighScore Plus software [37] with the International Centre for Diffraction Data (ICDD) PDF-2 database [38].

At 7 and 28 days of curing, MIP samples with a thickness of 2 mm were prepared using a precision saw. The specimens were immersed in isopropanol to stop further hydration until the MIP tests was conducted. After vacuum drying of the samples, MIP (AutoPore IV 9500, Micromeritics, Norcross, GA, USA) was performed.

3. Results and Discussion

3.1. Free and Bound Cl Contents

Figure 3 shows the measured contents of free and bound chloride in the FSW and SSW samples. In this case, the bound chloride content of the FSW sample increased from 59% to 69%, while that of the SSW sample did from 42% to 91% between 7 and 28 days. Although SSW did not show any noticeable advantage in immobilizing free chloride, compared to FSW, at 7 days, it clearly did at 28 days.



Figure 3. Relative % contents of free and bound chloride in the FSW and SSW samples at 7 and 28 days.

3.2. XRD Results

Figure 4 presents the XRD results of the hardened FSW and SSW samples and identified phases at 7 and 28 days.

Before discussing the phase identifications in detail (see Sections 3.2.1 and 3.2.2), it is worth noting that two major binding mechanisms for chloride in hydrated portland cements are known: (1) the formation of Cl-bearing Al₂O₃-Fe₂O₃-mono (AFm) phases, such as Friedel's salt and Kuzel's salt, and (2) surface-sorption by calcium silicate hydrate (C-S-H) [39]. Among these two mechanisms, Hosokawa reported that the chemical binding by AFm phases more greatly contributed to chloride uptake than the physical binding by C-S-H surfaces [40], indicating that the chemical binding by Cl-bearing phases would be an effective way to increase the Cl-binding capacity of hydrated portland cements. Therefore, similarly, the chloride-binding capabilities of the FSW and SSW samples would be importantly affected by the formation of Cl-bearing phases.



Figure 4. Cont.



Figure 4. XRD patterns of (**a**) raw fly ash and FSW, and (**b**) raw GGBFS and SSW samples at 7 and 28 days; dashed line 'S' is the location of the (0 1 1) basal reflection of Cl-sodalite; dashed line 'C' is the location of the (0 0 3) basal reflection of Cl-Chabazite.

3.2.1. XRD for the FSW Sample

In Figure 4a, the FSW sample contains original phases of raw fly ash (e.g., quartz, mullite, hematite, and magnesioferrite) that have been often observed in alkali-activated fly ash binders [11,41,42].

As reaction products, the FSW sample produced geopolymer, which manifests itself in the form of an amorphous hump at 28–35° in XRD [12,13,43], with several crystalline zeolite phases. At 7 days, besides geopolymeric gel, zeolite Y (Al₇₁Na₇₁O₃₈₄Si₁₂₁) and Cl-sodalite (Al₆Cl₂Na₈O₂₄Si₆) are only visible. At 28 days, while zeolite Y was more synthesized, zeolite X (Al₉₂Ca₃₂K₂₈O₃₈₄Si₁₀₀) and Cl-chabazite (Al₂Ca₁Cl₂O₈Si₂) are newly grown.

Zeolites X and Y are very similar zeolites that belong to faujasite. Their main difference is simply the Si/Al ratio [44]. It is worth noting that Belviso et al. synthesized zeolites from NaOH-activated fly ash using seawater within the temperature range of 35 to 60 °C [21]; in their study, zeolite X was

observed when the samples were prepared with seawater at 35 °C, while it was not found in the samples prepared with a distilled water at the identical temperature. This result implies that zeolite X is a more stable phase in seawater than in distilled water at the atmospheric temperature. Similarly, in the present study, the active formation of zeolite X was observed along with zeolite Y. However, zeolites X and Y were not likely to contribute to the chemical Cl-binding because they contain no chlorine in their chemical compositions.

In the FSW sample, the Cl-bearing phases were Cl-chabazite and Cl-sodalite, as shown in Figure 4a. These phases are the resultant zeolites after binding the chloride. Note that geopolymeric gel resembles zeolites in terms of structure on the nanometer scale [13,31], and thus the presence of Cl-chabazite and Cl-sodalite may indicate that a portion of the geopolymeric gel in the FSW sample is very similar to them in terms of structure and composition. Therefore, these types of geopolymeric gels are likely to chemically hold a portion of the chloride in the FSW sample. However, given that the XRD peaks for zeolites X and Y are much larger than those of Cl-chabazite and Cl-sodalite, the zeolite X and Y types of geopolymer gel seem to have a more significant portion in the FSW sample. This may explain the limited degree of chloride binding in the FSW sample as shown in Figure 3.

3.2.2. XRD for the SSW Sample

Figure 4b presents the XRD patterns of the SSW sample at 7 and 28 days. Here, calcium silicate hydrate I (C-S-H(I)), which is a more crystalline form of C-S-H, is found as a main reaction product [31,45]. Earlier studies have showed that C-S-H in concrete has the capacity to adsorb chloride ions on its surface area, referred as physical Cl-binding in hydrated portland cements [46–48]. Although the peaks of C-S-H(I) have not much grown after 7 days, C-S-H(I) would adsorb chloride to some extent.

In XRD, Cl-hydrocalumite and Cl-hydrotalcite are also identified. In hydrated portland cements, AFm phases, which belong to layered double hydroxides (LDH), play important roles in chloride chemical binding. In general, Mg, Ca, and Al sources in raw GGBFS lead to the formation of hydrotalcite (Mg-Al LDH) and hydrocalumite (Ca-Al LDH) in alkali-activated GGBFS [39]. Both hydrotalcite and hydrocalumite have LDH structures with interlayered anions, which can be exchanged with chloride ions [40,49]. In the present study, Figure 4b illustrates the formation of Cl-exchanged hydrotalcite (i.e., Cl-hydrotalcite) and Cl-exchanged hydrocalumite (i.e., Cl-hydrocalumite) in the SSW sample. Khan et al. recently found Friedel's salt as a chloride "sink" in four M-NaOH-activated GGBFS prepared with 2.47% NaCl solution [22]. Although Friedel's salt and Cl-hydrocalumite are often mistaken as an identical phase [50,51], their XRD patterns [38] are not the same to each other even though these phases have a great similarity in their structures (i.e., LDH); Cl-hydrocalumite contains chloride as a dominantly interlayered anion with a small quantity of OH ions in its interlayer spaces, while Friedel's salt has only Cl ions. Due to their structural and compositional similarities, Cl-hydrocalumite may have a similar binding capability for chloride. However, as shown in Figure 5, which is the detailed XRD figure that shows the strongest peaks of Cl-hydrotalcite and Cl-hydrocalumite in 8° to 15° 20 [38], only Cl-hydrotalcite seems to have significantly grown in its peak intensity from 7 to 28 days compared to Cl-hydrocalumite. Given that the bound chloride content in the SSW sample considerably increased from 7 to 28 days (Figure 3), Cl-hydrotalcite would play a more important role in binding chloride rather than Cl-hydrocalumite in this study.



Figure 5. Detailed XRD figure of the SSW sample in 8–15° 20 at 7 and 28 days with reference patterns of Cl-hydrocalumite and Cl-hydrotalcite.

3.3. MIP

Figure 6 (see also appendix Table A1) show the MIP results of the FSW and SSW samples at 7 and 28 days.

As shown in Figure 6a, the FSW sample at 7 and 28 days has similar values of total porosity (i.e., an insignificant reduction of total porosity). However, in Figure 6b, the pore-size distributions display notable differences between 7 and 28 days in two size ranges: (1) 200–500 nm, and (2) below 30 nm. Pores with sizes of 200–500 nm have decreased while pores smaller than 30 nm increased from 7 to 28 days. Thus, the FSW sample had a pore-size refinement.

In contrast, the SSW sample demonstrates a great reduction in total porosity from 0.08 to 0.03 mL/g, as shown in Figure 6c. The SSW sample also shows a substantial pore-size refinement (see Figure 6d). For instance, the pores larger than 0.1 μ m at 7 days mostly disappeared at 28 days, while the pores smaller than 0.005 μ m increased. The simultaneous reduction of pore size and total volume are advantageous to improve impermeability to chloride ingression.



10 of 14

Figure 6. Cont.



Figure 6. Pore size distribution curves of the FSW and SSW samples at 7 and 28 days: (**a**) cumulative distribution of the FSW sample, (**b**) log differential distribution of the FSW sample, (**c**) cumulative distribution of the SSW sample, and (**d**) log differential distribution of the SSW sample.

3.4. Summary and Further Discussion

In the present study, the alkali-activated slag system revealed a better Cl-binding capacity than the alkali-activated fly ash system after 7 days, although they had similar capacities at 7 days. In the alkali-activated slag system, Cl-hydrotalcite remarkably formed after 7 days compared to Cl-hydrocalumite, which has a similar structure with Friedel's salt. Therefore, the chemical Cl-binding mechanism in the alkali-activated slag is considerably different from the hydrated portland cement system, of which the chemical Cl-binding mostly depends on Friedel's salt.

In the alkali-activated fly ash system, the Cl-bearing zeolites did not extensively form after 7 days as much as Cl-hydrotalcite in the alkali-activated slag system; however, zeolite X and Y were substantially synthesized after 7 days. Given that, in earlier studies [12,13,31], when no Cl was present, dominant crystalline zeolite phases in the alkali-activated fly ash system were mostly sodalite and chabazite rather than faujasite, it is possible that the extensive formation of zeolite X and Y might suppress the formation Cl-bearing zeolites in this study, although a further study is required to understand the interactions between zeolite X/Y and Cl-bearing zeolites in the system.

Regarding the pore structures, the alkali-activated slag system revealed both pore-size refinement and porosity reduction after 7 days, while the alkali-activated fly ash system had only pore-size refinement. Therefore, considering Cl-binding and pore structure, the alkali-activated slag system would have a better durability performance in seawater environments than the alkali-activated fly ash system.

4. Conclusions

The present study compared and discussed the chloride-binding capability, synthesized Cl-bearing phases, and pore characteristics between alkali-activated fly ash and GGBF systems when seawater is used as the mixing water. Overall, the alkali-activated GGBFS system showed a significantly better Cl-binding capacity than the alkali-activated fly ash system at 28 days although it demonstrated a slightly lower performance at 7 days. More detailed conclusions are given as follows:

- 1. The main Cl-bearing phases were zeolites (i.e., Cl-chabazite and Cl-sodalite) in the FSW sample, while they were LDH phases (i.e., Cl-hydrotalcite and Cl-hydrocalumite) in the SSW sample.
- 2. The bound chloride content of the SSW sample was 91% at 28 days, whereas that of the FSW sample was 69%. Therefore, at 28 days, the alkali-activated GGBFS system is more advantageous for Cl-binding than the alkali-activated fly ash system.

- 3. The FSW sample contained zeolite X, zeolite Y, Cl-chabazite, and Cl-sodalite as crystalline zeolite phases. Zeolite X and Y, which do not contribute to the chemical Cl-binding, comprised a larger portion of the XRD pattern than Cl-chabazite and Cl-hydrocalumite.
- 4. The SSW sample contained C-S-H(I), Cl-hydrocalumite, and Cl-hydrotalcite. Its Cl-binding mechanisms are similar to hydrated portland cement systems. C-S-H(I) is responsible for physical binding and Cl-LDHs contribute to chemical binding.
- 5. Between the two Cl-LDHs in the SSW sample, the XRD peak of only Cl-hydrotalcite grew significantly from 7 to 28 days, and the bound chloride content of the SSW sample considerably increased from 42% to 91% over that time. Therefore, after 7 days, Cl-hydrotalcite is an important phase for chemical Cl-binding in the alkali-activated GGBFS system.
- 6. The SSW sample showed both pore-size refinement and porosity reduction from 7 to 28 days. In contrast, the FSW sample only showed pore-size refinement over that time.

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Conflicts of Interest: The authors declare no conflict of interest.

Appendix A

Pore size Diameter (µm)	Log Differential Intrusion (mL/g)		Pore size Diameter	Log Differential Intrusion (mL/g)	
	FSW 7 Days	FSW 28 Days	(μm)	SSW 7 Days	SSW 28 Days
0.00329	0	0.12304	0.00303	0.12477	0.03992
0.00362	0	0.14457	0.00329	0.11614	0.24018
0.00402	0	0.11852	0.00362	0.07304	0.12966
0.00452	0	0.0805	0.00402	0	0.06723
0.00517	0	0.06687	0.00452	0	0
0.00603	0	0.07686	0.00517	0	0
0.00724	0	0.06765	0.00603	0	0
0.00906	0	0.06558	0.00724	0.00464	0
0.01105	0	0.05297	0.00906	0.00489	0
0.01224	0	0.05159	0.01105	0.01095	0
0.01373	0.01259	0.04108	0.01224	0.00332	0
0.0171	0	0.03531	0.01373	0	0
0.02108	0.00507	0.02946	0.0171	0	0
0.02629	0.01781	0.03803	0.02108	0	0
0.03239	0	0.03331	0.02629	0.0025	0
0.04033	0.07809	0.05043	0.03241	0	0
0.05034	0.02662	0.04926	0.04032	0	0
0.06249	0.05212	0.05721	0.05035	0	0
0.0771	0.05404	0.0631	0.0625	0.00274	0
0.09539	0.06335	0.07466	0.07709	0	0
0.12075	0.08667	0.08664	0.09537	0.00672	0
0.15107	0.09919	0.1101	0.12082	0.01236	0

Table A1. The detailed pore volumes of mercury intrusion porosimetry (MIP) results.

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